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TRANSLATOR'S AFFIDAVIT

I, Herbert Dubno, a citizen of the United States of America, residing in Bronx (Riverdale), New York, depose and state that:

I am familiar with the English and German languages;

I have read a copy of the German-language document attached hereto, namely PCT/DE03/01484; and

The hereto-attached English-language text is an accurate translation of the above-identified German-language document.

Herbert Dubno

Sworn to and subscribed before me
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T R A N S L A T I O N

D e s c r i p t i o n

METHOD FOR PRODUCING HIGHLY POROUS METALLIC MOLDED BODIES TO CLOSE TO A FINAL CONTOUR

The invention relates to a process by means of which porous and especially highly porous components can be produced to close to a final contour.

State of the Art

The pressing of metal powders for the production of porous metal bodies is known. To produce the desired porosity the so-called place-holder material [dummy material] can be added to the metal powder to enable the desired porosity to be stabilized. After pressing of the green body from the powder mixture, the place holder material is then removed from the green body so that the green body consists only of the remaining metal powder framework which has spaces within its framework structure. The green body has thus already the porous structure which is later to be found in

the molded body. In the driving off of the place-holder material, one must be concerned to maintain the metal powder framework. By means of the subsequent sintering of the base body, a high porosity molded body can be obtained in which the powder particles are diffusion bonded together at their contact surfaces by sintering.

As the place-holder material [dummy material] for the formation of porous metallic molded bodies, relatively high melting organic components are known which by vaporization [evaporation] or pyrolysis (cracking) and the solubilization of the resulting product by means of appropriate solvents can be removed from the green bodies. It is a problem with such materials that significant time is cost by the removal of place-holder materials and cracking products which can react with practically all of the metals used in powder metallurgical processes like titanium, aluminum, iron, chromium, nickel, etc. so that high concentrations of impurities remain. It is also a disadvantage where thermoplasts, which are to be removed by heating the green body, are used in that the expansion at the glass transition point has a detrimental effect on the requisite stability of the green body.

Alternatively, high melting inorganic contents, like alkali salts and low melting metals like magnesium, tin, lead, etc. are also used as place holders [dummy materials]. Such place holder materials are removed in vacuum, or under a protective gas at temperatures between about 600 to 1000°C from green bodies at

high energy cost and in a time-consuming manner. It cannot be avoided with such place-holder materials that impurities will remain in the green body which may be detrimental especially in the case of molded bodies of reactive metal powders like titanium, aluminum, iron, chromium and nickel.

From DE 196 38 927 C2, a method of making highly porous metallic molded bodies is known in which initially metal powder and a place holder are mixed and then pressed to a green mass. In this operation both uniaxial as well as isostatic pressing can be used. The place holder or dummy is then thermally driven out and the green body then sintered. If the powder-dummy mixture is stabilized with a binder, it is in principle possible to produce even relatively complex component geometries by multiaxial pressing. The fabrication of the pressing dies for this purpose is however expensive and difficult. Especially for small series of pieces it is therefore advantageous to produce semifinished products or blanks with a universal geometry (for example cylinders or plates) and then by subsequent mechanical processing to impart the desired final contour to the product.

According to the present state of the art, the final shape is imparted to highly porous shaped bodies only after the sintering by conventional mechanical methods like for example turning, milling, boring or grinding. It is a disadvantage of these subsequent machining operations that the already sintered

blank is connected with a local workpiece deformation. Through the plastic deformation there is usually a smearing of the pores. As a consequence the desired open porosity of the molded body is generally lost precisely in those surface regions at which it is desirable. This has a detrimental effect on the functional characteristics of the molded body. Furthermore, the workpiece, because of its porosity can only be clamped and machined with great care since it is not very stable under compression. The nonuniform surface of the porous molded body gives rise to a relatively high tool wear.

Object and Solution

The object of the invention is to provide a simple method of making a high porosity metallic shaped body which can have an especially highly complex geometry, which is free from the aforedescribed drawbacks like the detrimental effect on the porosity at the surface.

Subject of the Invention

The subject of the invention is a method of making high porosity metallic shaped bodies. The method thus comprises the following method steps: A metal powder to be used as a starting material is mixed with a place holder or dummy. The metal powder can be, for example, titanium and its alloys, iron and its alloys,

nickel and its alloys, copper, bronze, molybdenum, niobium, tantalum or tungsten.

The materials suitable as place holders or dummies are for example carbamide $CH_4N_2O(H_2N-CO-NH_2)$, biuret $C_2H_5N_3O_2$, melamine $C_3H_6N_6$, melamine resin, ammonium carbonate $(HN_4)CO_3H_2O$ and ammonium bicarbonate NH_4HCO_3, which can be removed without leaving residue at temperatures of up to 300°C from the green body. Especially advantageous as the place holder material or dummy is ammonium-bicarbonate which can be driven out into the air already at about 65°C. The grain size, that is the particle size, and the particle shape of the place-holder material or dummy determines the porosity to be formed in the molded body. Typical particle diameters of the place holder material or dummy are 50 μm to 2 mm. By suitable choice of the place holder or dummy and the amount of the place holder or dummy with respect to the metal powder, a high, homogeneous and open porosity can be produced in the final molded body. Porosities of up to 90% are achievable without more.

From the mixture a green body, especially a green body with a simple geometry, is pressed. The green body can for example by a cylinder or also a plate. The press process can use multiaxial pressing or cold isostatic pressing. The multiaxial pressing results in a dimensionally stable semiproduct or blank with a defined external contour. The wall friction and demolding results in the formation of a so-called press skin which is formed

from plastically deformed metallic particles. This press skin can be removed prior to sintering by mechanical machining to the extent no further green machining is required. The wall friction limits the length-to-diameter ratio to 2:1. Above this value density differences in the pressed body which are too great arise. The cold isostatic pressing is carried out for example in rubber molds. As the pressure transmission medium, an oil-containing emulsion can be used in which the powder filled rubber mold is immersed. Since the wall friction on demolding is thereby eliminated, it is possible to make blanks with a length to diameter ratio greater than 2:1 and with a sufficiently homogeneous density distribution. It is a drawback that the dimensional stability of the outer contour is somewhat limited although this has scarcely any effect on the subsequent green processing.

The green body is then subjected to a conventional mechanical machining in which the workpiece is provided with its final form, with the shrinkage during the sintering process being calculated in. The machining is done in the green state in which the mass still contains the place holder or dummy, with the advantage that the workpiece can be machined very simply and the porosity is not affected. The tool wear is then usually held low. Even highly complex shapes can be imparted with this process. The still present place holder or dummy makes the workpiece to be machined sufficiently stable against compression to enable it to be clamped for the subsequent mechanical machining.

When the final shape has been produced, the plate holder material is removed in air or under vacuum or under a protective gas from the green body thermally. The atmosphere which is used is dependent upon the place holder or dummy material which is selected. For example, air as an atmosphere suffices for the removal of ammonium bicarbonate as the place holder or dummy at a temperature above 65°C. The green body is then sintered to produce the molded product.

The mechanical machining prior to sintering advantageously enables simple production of a molded body close to the final contour even for complicated geometry of the molded body to be produced without detriment to the porosity and without high tool wear.

This process is not limited only to the production of molded bodies with a unitary porosity but it allows for the production of molded bodies with different porosities, for example, graded porosity.

In the use of coarse starting powders generally the single particles have only a weak connection to the sintered network since the sintered bridges are only incomplete. Even with small loads, such bodies generally can break down. This can however be impermissible for certain applications. In order to avoid this detrimental effect, high porosity components from coarse

starting powders before use are advantageously trovalized or ground smooth. In this process the weakly adherent particles are usually removed by a grinding step from the surface.

Special Description Part

In the following the subject of the invention is described in greater detail in connection with the Figures and an example without thereby limiting the subject of the invention.

The drawing shows:

FIG. 1: possible embodiments of the semifinished product or blank which are produced by multiaxial pressing and by cold isostatic pressing.

FIG. 2: different metal geometries which are made from stainless steel 1.4404 (316L) by the process according to the invention.

FIG. 3: an illustration of the microporosity which is set by the place holder or dummy material and the microporosity within the sintered webs.

The typical method steps for a method according to the invention are as follows:

1. Initially the blank is made as described in DE 196 38 927. For that purpose metal powder, especially stainless steel 1.4404 (316L) or titanium is mixed with a place holder or dummy, especially ammonium bicarbonate and uniaxially or cold isostatically pressed. The blank, for example a cylinder or a plate, as required for further processing is made with a suitable die. FIG. 1 shows possible embodiments of the blank which are made by multiaxial pressing and by cold isostatic pressing.

2. There follows the green machining of the unsintered blank by conventional mechanical machining operations (sawing, boring, turning, milling, grinding ...). The place holder or dummy advantageously increases the green strength of the blank and thus has a positive effect on the machinability. A further advantage of the machining is the low cutting force and thus the limited tool wear. A smearing of the pores is also avoided.

3. The removal of the place holder or dummy and the sintering can be carried out conventionally on a planar sintering surface of ceramic or alternatively in a bed with ceramic balls. The parameters of the removal of the place holder or dummy can be those of DE 196 38 927 C2.

As a complement to DE 196 38 927 C2, it can be noted that the removal of the place holders ammonium carbonate and ammonium bicarbonate can take place in air. The sintering in a ball bed has

the advantage that the contact surfaces against the component are limited so that an adhesion of the components to the ceramic balls is prevented. The ball bed easily compensates for the sintering shrinkage by the reorientation of the balls so that a uniform contact with the sintering surface is ensured during the entire sintering process. This avoids distortion of the components made during sintering. As an option the molded body, to improve the surface quality, can then be trovalized.

Exemplary Embodiment

FIG. 2 shows different metal geometries which are made from the stainless steel 1.4404 (316L) according to the invention and with the method sequence described in the following. As the starting material a water-atomized powder (grain fraction below 500 μm) was used. The steel powder was mixed with the place holder or dummy ammonium bicarbonate (grain fraction 355 to 500 μm) in a ratio of steel powder to ammonium bicarbonate of 45 to 55 (in volume %). This corresponded to a ratio of steel powder to place holder of 80.5 to 19.5 in weight %. The mixture was uniaxially pressed with a press pressure of 425 MPa to cylinders with a diameter of 30 mm and a height of 22 mm. The cylinders were machined in the green state by turning and drilling. Apart from bores the cylinders can also be provided with right angled and also rounded shoulders in the model geometry. The removal of the place holder ammonium bicarbonate was effected in air at a temperature of 105°C. The decomposition of the place holder or dummy occurred

already at 65°C but the higher temperature was chosen to drive off the decomposition product water in the gaseous state. The sintering was carried out at 1120°C for two hours under an argon atmosphere. The metal geometry showed a shrinkage of about 4%. The final porosity of the fabricated component was about 60%. It was a result of both the macro porosity established by the place holder material and the micro porosity which developed in the sintered web (FIG. 3). The micro porosity resulted from incomplete sintering of the metal particles. A reduction of the micro porosity could be obtained by the use of finer starting powders or by sintering at higher temperatures.